



UNIVERSIDADE ESTADUAL DE CAMPINAS SISTEMA DE BIBLIOTECAS DA UNICAMP REPOSITÓRIO DA PRODUÇÃO CIENTIFICA E INTELECTUAL DA UNICAMP

Versão do arquivo anexado / Version of attached file:

Versão do Editor / Published Version

Mais informações no site da editora / Further information on publisher's website: https://www.scielo.br/scielo.php?script=sci_arttext&pid=S1517-70762017000400416

DOI: 10.1590/s1517-707620170004.0227

Direitos autorais / Publisher's copyright statement:

©2017 by UFRJ/Coordenação dos Programas de Pós-Graduação de Engenharia. All rights reserved.

DIRETORIA DE TRATAMENTO DA INFORMAÇÃO

Cidade Universitária Zeferino Vaz Barão Geraldo CEP 13083-970 – Campinas SP Fone: (19) 3521-6493 http://www.repositorio.unicamp.br



Microstructural characterization of the A356 alloy in different processing routes used in the thixoforming process

Luis Vanderlei Torres^{1, 2}, Luis Fernando Torres², Eugênio José Zoqui²

V 77 N NA

¹Federal Institute of Education, Science and Technology of São Paulo – IFSP PC: 12929-600, Bragança Paulista, SP e-mail: torres@ifsp.edu.br

Materials and Manufacturing Engineering Department of Faculty of Mechanical Engineering of University of Campinas - UNICAMP PC: 13083-860, Campinas, SP

e-mail: torres@fem.unicamp.br; lftorres00@yahoo.com.br; zoqui@fem.unicamp.br

ABSTRACT

This work aims to evaluate different routes of raw material production to be used in thixoforming process. The semisolid materials exhibit particular structural characteristics, such the characteristic non dendritic structure. The studies for the production of near-net-shape parts have progressed together with the studies of material weight reduction on the automobile industries production lines. In general the advantages are energy saving, manufacturing steps reduction, productivity and quality increasing. Five processing routes have been investigated (conventional casting, electromagnetic stirring, grain refinement technique and different cooling rates) through microstructural characterizations, conventional metallographic and colour metallography. The material used in this study was the A356 alloy (commercial) having a hypoeutectic structure and grain size of about 750 µm and primary dendrite arm spacing size of around 80 µm, the combination of electromagnetic stirring + grain refining + high cooling rate provided the best route processing to obtain a refined structure, with grain sizes of around 240 µm, being feasible at the thixoforming process. Note also that using conventional metallography (black and white) all routes tested showed almost the same primary dendrite arm spacing size values ie had identical characteristics. Importantly, only with the characterization via polarized colour metallography is that was achieved in fact the best processing route definition.

Keywords: A356 alloy, casting, electromagnetic stirring, grain refinement, thixoforming.

1. NTRODUCTION

Thixoforming is a technological process that involves the metal alloys forming in the semisolid state (SSM). Interest in this process is justified by the numerous advantages that can be obtained compared to the conventional manufacturing methods: smaller turbulence during filling matrices, smaller shrinkage during solidification, longer dies and tooling lifetime, smaller loads involved, among others. The semisolid processing of metals has been widely accepted by the industry as a feasible route to the of near net-shape components of aluminum and magnesium alloy production [1,2]. The electronics, aerospace and especially the automobile industries already use components manufactured by such processing. In Europe, it highlights parts such as suspension parts, brackets for motors and pipes to fuel injection for industries such as Alfa Romeo, Fiat, Peugeot and Renault and already in the USA, the production includes bikes and snowmobiles mechanical components, while in Asia, there is a concentration in the production of electronic components and power supply, particularly using magnesium alloys [3,4].

Aluminum castings alloys have a fundamental role in the metal-mechanics industry, these alloys are supplied in a wide range of chemical compositions [5,6]. Among the various possibilities of thixoformable materials, Al-Si alloys are the most frequently used, particularly A357 and A356 alloys present favourable thermodynamic characteristics to thixoforming: at the eutectic temperature, liquid fraction is around 50% and the 50% of solid is the primary alpha phase. In this condition, the material thixoformability depends on the morphology and the solid crystals size present in the semisolid: small dimensions and globular morphology are required [7]. Raw material production to thixoforming usually involves high cost equipments; the main purpose of this work is to investigate low cost procedures to allow wider popularization of the SSM processing. Cast structures with non-dendritic, small equiaxial grains can be produced using different approaches: chemical agents (appropriate choice of alloying elements or grain refiners) and physical or thermal agents (mechanical/electromagnetic stirring, vibration, low pouring temperature, high cooling rate, etc) [2,7,8,9]. The use of the electromagnetic stirring technique has been widely used in the last years for the thixoforming processes raw material production due to its numerous advantages, among them we can mention the absence of contact between the liquid metal and the stirring environment, fact that doesn't happen in the mechanical agitation, allows the continuous and direct casting of billets with varied forms, has low electric energy consumption, among others. Electromagnetic stirring is promoted in the solidifying liquid by the action of strong electromagnetic fields. The field-induced electric currents promote a strong agitation, breaking the structure in formation and stimulating the crystalline multiplication; achieving low values of primary globules sizes and grains that are important factors for the thixoforming process. [10,11,12,13]. In order to, this work aims to evaluate different processing routes, using A356 alloy as base material the in terms of microstructural characterization by optical microscopy, using conventional metallographic (black and white) and polarized metallography (colour).

2. MATERIALS AND METHODS

The material used in this work was the commercial A356 alloy whose chemical composition can be seen in Table 1. To the alloy chemical composition analysis via optical emission spectrometry equipment, the samples were abraded with water sandpaper grain size of 220, 320, 400 and 600 mesh and were finished for about 30 seconds in ultrasound to remove any impurities. The material was acquired in the market in cast condition (gravity casting).

Table 1: A356 alloy chemical composition (in wt %): *obtained in the literature; **chemical analysis performed by this foundry industry commercial; ***and obtained by optical emission spectrometry;.

	Si	Mg	Fe	Cu	Zn	Mn	Ti	AI
*A356 Min - Max [14]	6,50 - 7,50	0,25 - 0,45	0 - 0,20	0 - 0,20	0 - 0,10	0 - 0,10	0 - 0,20	Bal.
**A356 (by industry)	6,72	0,38	0,20	0,09	0,02	<0,01	<0,01	Bal.
***A356	6,96	0,38	0,19	0,08	0,02	<0,01	<0,01	Bal.

The characterization of the solidus and liquidus temperatures of the alloy is given by simulation software Thermo-Calc® through the chemical composition of the analyzed alloy by optical emission spectrometry, the solidus temperature is 557 °C and the liquidus temperature is 612 °C, namely, a semisolid temperature range of 55 °C. It is observed by solidification path shown in Figure 1 two eutectic temperatures (TE) 577°C and 557°C. The first TE (577°C) is the solidification of the second phase (silicon particle) and second TE (557°C) represents the solidification of the third phase (Mg₂Si intermetallic phase), finally forming a interdendritic eutectic mixture composed of Al- α +Si+Mg₂Si. The proper characterization of this temperature is extremely important since this eutectic to liquefy increases the material temperature quickly which could compromise the fusion or solidification structure control during the processing. A casting temperature of 50 °C above the liquidus temperature was used because of the temperature drop associated with the furnace raw material withdrawal moment until its casting moment.



Figure 1: Mass fraction of liquid versus temperature for A356 alloy predicted by software Thermo-Calc® showing the present transformations (database: TTAI5).

The alloy was melt and poured at 660 °C in a Cu mould as shows the Figure 2, producing up ingots with approximately 250 mm in length and 30 mm in diameter. The first processing technique was used as comparison parameter, since it is a conventional casting, without electromagnetic stirring and without grain refiner use. The use of electromagnetic stirring [10,11,12,13] has great benefits to obtain refined structures, because promotes the breakdown of structure formation, stimulating like this crystalline multiplication and its consequent globularization. In the whole tests which utilized electromagnetic agitation was used the higher power provided by the system, 8kW that is enough to produce a magnetic field of 13 Gauss [12]. The used water flow rate for cooling the metal mould was given using the two extremes provided by the system, a minimum flow rate of 1 l/min and a maximum flow rate of 10 l/min.



Figure 2: Schematic representation of the casting assembly to produce ingots under different conditions: 1) Cu mould, 2) electromagnetic coils - 13 Gauss, 3) electric motor, 4) water drain and 5) water inlet [18].

In this work were investigated five different processing routes aiming to optimize the process, obtaining a refined structure. The different processing routes of the raw material are summarized in Table 2.

CONDITIONS	DESCRIPTIONS		
Casting	casting in Cu mould + high water cooling (10 l/min)		
EMS + HWC	electromagnetic stirring + high water cooling (10 l/min)		
EMS + LWC	electromagnetic stirring + low water cooling (1 l/min)		
EMS + GR + HWC	ectromagnetic stirring + grain refiner + high water cooling (10 l/min)		
EMS + GR + LWC	electromagnetic stirring + grain refiner + low water cooling (1 l/min)		

Table 2: Summary of the different processing routes investigated.

The samples produced were characterized by optical microscopy to evaluate the morphological changes resulting from several investigated conditions, all the samples were submitted to conventional metallography (black and white) and polarized metallography (colour). Samples were cut longitudinally in the middle of the ingot, whose size was 300 mm, in sequence the samples were sanded with sandpaper in a particle size of 220, 320, 400, 600, 800, 1200 and 1500 mesh and polished with diamond paste of 1µm. After polishing, the samples were attacked with hydrofluoric acid (1 ml HF and 99ml H₂O), the attack was performed with total immersion of the sample in the reagent during 10s. To the microstructural characterization by colour metallography, the same samples used in previous characterization passed through an electrolytic attack with deposition of HBF₄ (hydrofluoric acid) in 2,0% solution and 25V voltage for about 6 minutes under moderate and constant stirring. To the image acquisition we used a optical microscope Leica DM ILM. Polarizing filters have been used to obtain the samples grains colour images, doing the grains that have the same crystal orientation exhibit similar colour, which makes it much easier its identification and characterization thereof. To the primary dendrite arm spacing size measurements (λ_1) in conventional metallography and grain size in colour metallography we used Heyn intercepts method. The count of the primary dendrite arm spacing was made by building up a triad of length equal to a multiple of full scale used in the image, counting the number of intersections covered by each straight line forming the triad. After the first count the triad was moved to the other four different positions, thus five different count fields were analyzed in each micrograph, four images being used in different regions of the sample for a total of 20 fields analyzed for each condition. With the obtained data was generated an average primary dendrite spacing of the four counting micrographs obtained. With such value was hand picked the size of the line forming the triad, and its length compared to the micrograph scale line length, through a three-rule, it was calculated the triad line micrometers size. Finally, dividing the length in micrometers of a triad line the average primary dendrite spacing count, an average size of primary dendrite spacing was obtained.

3. RESULTS AND DISCUSSION

Figure 3 shows the micrographs of the A356 alloy (commercial), it is observed that the alloy has primary dendrite arm spacing size much higher as compared to the investigated conditions; as well as a microstructure in the form of gross dendrite, typical commercial foundry. When using images got by polarized light, we have that the same showed efficient to determine the grain sizes. Thus, the colour metallography wich can be seen is a quite high grain size, of about 750 µm and a dendritic microstructure, showing that the use of such material to the thixoforming process is not feasible.



Figure 3: A356 alloy (commercial) - (a) Conventional B&W metallography and (b) Polarized colour metallography.

Figure 4 shows the micrographs of different processing routes investigated by conventional B&W metallography and polarized colour metallography. In the conventional B&W metallography notice a typical hypoeutectic structure as a result of solidification, there is dendrite of the primary phase (α -Al) and a network of the interdendritic constituted for the Al-Si eutectic. The dendritic growth is evident, however, due to the high solidification rate imposed by the used metal mould (cooling) as well as the electromagnetic stirring, the primary dendrite arm spacing is relatively small, in order of 30 µm, there was a large reduction when compared with the material (commercial) it is known that the grain refining technique for Al-Si-Cu alloys with low silicon percentages (1wt% Si, 2wt% Si and 4wt% Si) is effective, but for high silicon percentages, in the case of 7wt% Si (similar to the A356 alloy, but with the addition of 2.5 wt% of Cu), it doesn't have a satisfactory result, presenting a totally dendritic structure [15]. Knowing that the grain refining technique for alloys with high silicon percentages is not effective, the electromagnetic stirring technique was added to the material's production, in according to EMS + GR + HWC and EMS + GR + LWC conditions. Studies that report only the use of electromagnetic agitation show the benefits of obtaining refined structures, providing primary globules and grains considerably inferior when compared to non-agitated structures [16,17]. When used grain refiner (Al-5wt%Ti-1wt%B) notices a change in its shape, namely, the material passes from a typically dendritic structure to a structure trending to the rosette shape, with a substantial gross on the dendritic branches (EMS + GR + HWC and EMS + GR + LWC).

The micrographs with polarized colour metallography revealed relatively gross equiaxed grains, with an average size of 400 μ m, however when using the combination electromagnetic stirring and grain refining technique (EMS + GR + HWC/LWC), the average size falls substantially, namely, 240 μ m in other words, the combination of these two production techniques is the one that presents the best results in relation to grain size, an important factor when dealing with semi-solid materials processes [18,19,20]. Therefore, the use of the grain refining technique interferes with nucleation, causing an increase in the initial nucleation rate, that is the number of nuclei as a function of time and volume of the liquid, thus restricting its growth and resulting in a solid structure of fine grains. It is therefore important to encourage extensive nucleation throughout the liquid volume when associated with high cooling rates [21].Importantly, only with the characterization via polarized colour metallography is that can be obtained in fact the best condition.





Figure 4: Micrographs of the different processing routes investigated by conventional B&W metallography and polarized colour metallography (samples were cut longitudinally in the middle of the ingot).

On the Table 3 the average size of primary dendrite arm spacing and average grain sizes to all processing routes investigated are presented. Note the combination of electromagnetic stirring + grain refining technique + high flow rate (EMS + GR + HWC) provides the best values on the average grain size, since: the grain refining acts as a catalyst nucleation; the agitation promoted on the liquid in solidification generates strong electromagnetic fields, contributing to the breakdown of the structure in formation and stimulating the crystalline multiplication, preventing dendritic growth. Which can be noted a gradual average grain size decrease in relation to the different investigated processing routes, especially when the use of electromagnetic stirring + grain refining technique + high water cooling, providing a grain size about 3 times smaller when compared to commercial alloy; it also notes that the primary dendrite arm spacing of different routes investigated remains almost the same and that only with the use of characterization via colorful metallography to the result be found [18].

CONDITIONS	PRIMARY DENDRITE ARM SPACING SIZE [µm]	GRAIN SIZE [µm]
Commercial alloy	85 ± 10	743 ± 258
Casting	31 ± 3	457 ± 125
EMS + HWC	31 ± 3	367 ± 106
EMS + LWC	31 ± 3	393 ± 155
EMS + GR + HWC	34 ± 3	244 ± 53
EMS + GR + LWC	36 ± 3	251 ± 52

 Table 3: Primary dendrite arm spacing size and grain size for each investigated condition.

4. CONCLUSION

Some conclusions about the different processing routes investigated in this work are:

- i. The A356 alloy processed through the combination of electromagnetic agitation + grain refining + high water cooling (EMS + GR + HWC) showed the best results for the grain size, demonstrating that this processing route is effective in thixoforming process, once from this process is expected homogeneous structures and with small grain sizes. Being that both techniques, electromagnetic stirring and ultra-refining technique provide the breakdown of the structures in solidification and prevent the raw material dendritic growth.
- ii. The microstructural characterization via colour metallography is quite efficient, doing with the grains with the same crystal orientation present similar colour, making it much easier identification and characterization thereof.

5. ACKNOWLEDGEMENTS

The authors acknowledge the Brazilian research funding agencies FAPESP (São Paulo Research Foundation – Project 2009/08478-1), CNPq (National Council for Scientific and Technological Development) and CAPES (Federal Agency for the Support and Evaluation of Postgraduate Education) for their financial support of this research. The authors are also indebted to FEM/UNICAMP – Faculty of Mechanical Engineering, State University of Campinas and IFSP – Federal Institute of Education, Science and Technology of São Paulo campus Bragança Paulista for its practical support.

6. BIBLIOGRAPHY

[1] HIRT, G., KOPP, R., Thixoforming: semi-solid metal processing, 1 ed., Aachen, WILEY-VCH, 2009.

[2] KIRKWOOD, D.H., SUÉRY, M., KAPRANOS, P., *et al.*, "Semi-solid processing of alloys", In: Springer Series in Material Science, v. 124, Berlin, Springer-Verlag, pp. 172, 2010.

[3] GARAT, M., MAENNER, L., SZTUR, C., "State of the art of thixocasting", In: *Proceedings of the 6th International Conference on the Semisolid Processing of Alloys and Composites*, pp. 187-194, Turin, 2000.

[4] YOUNG, K., EISEN, P., "SSM (semi-solid metal) technological alternatives for different applications", In: *Proceedings of the 6th International Conference on the Semisolid Processing of Alloys and Composites*, pp. 97-102, Turin, 2000.

[5] PERES, M.D., SIQUEIRA, C.A., GARCIA, A., "Macrostructural and microstructural development in Al-Si alloys directionally solidified under unsteady-state conditions", *Journal of Alloys and Compounds*, v. 381, pp. 168-181, 2004.

[6] COSTA T.A., MOREIRA, A.L., MOUTINHO, D.J., *et al.*, "Growth direction and Si alloying affecting directionally solidified structures of Al-Cu-Si alloys", *Materials Science and Technology*. v. 31, pp. 1103 - 1112, 2015.

[7] FLEMINGS, M.C., "Behavior of metals in the semisolid state", *Metallurgical Transactions A*, v. 22, pp. 957-981, 1991.

[8] SALLEH, M.S., OMAR, M.Z., SYARIF, J., et al., "An overview of semisolid processing of aluminium alloys", *ISRN Materials Science*, pp. 1-9, 2013.

[9] HAGA, T., NAKAMURA, R., KUMAI, S., et al., "Clad Strip Casting by a Twin Roll Caster", Archives of Materials Science and Engineering, v. 37/2, pp. 117-124, 2009.

[10] ZOQUI, E.J., PAES, M., ES-SADIQI, E., "Macro and microstructure analysis of the SSM A356 produced by electromagnetic stirring", *Journal of Materials Processing Technology*, v. 120, pp. 365-373, 2002.

[11] TORRES, L.V., ZOQUI, E.J., "Characterization of the microstructure and rheological behavior of Al-4wt%Si-2.5wt%Cu alloy produced by direct chill casting and electromagnetic stirring", *Solid State Phenomena*, v. 192-193, pp. 142-148, 2013.

[12] ZOQUI, E.J., CAMPO, K.N., PRONI, C.T.W., "Influence of the processing route on the microstructure of aluminum alloy A356", *Materials Characterization*, v. 85, pp. 26-37, 2013.

[13] TORRES, L.V., BENATI, D.M., ZOQUI, E.J., "Morphological evolution of Al-2wt%Si-2.5wt%Cu alloy produced by EMS and EMS enhanced by ECAP", *Solid State Phenomena*, v. 217-218, pp. 105-110, 2015.

[14] NADCA, Product specification standards for die castings produced by semi-solid and squeeze casting process, Wheeling, 2006.

[15] BENATI, D.M., Avaliação da tixoconformabilidade de ligas Al-Xwt%Si-2,5wt%Cu-0,5wt%Mg, Dissertação de M.Sc., FEM/UNICAMP, Campinas, SP, Brasil, 2008.

[16] PAES, M., Otimização de processo de obtenção de reofundidos da liga A356 por agitação eletromagnética, Dissertação de M.Sc., FEM/UNICAMP, Campinas, SP, Brasil, 2000.

[17] KANG, C.G., BAE, J.W., KIM, B.M. "The grain size control of A356 aluminum alloy by horizontal electromagnetic stirring for rheology forging", *Journal of Materials Processing Technology*, v. 187, pp. 344-348, 2007.

[18] TORRES, L.V., *Tixoconformação de novas ligas Al-Si-Cu*, Tese de D.Sc., FEM/UNICAMP, Campinas, SP, Brasil, 2013.

[19] VILAS BÔAS, B., Avaliação do comportamento mecânico de ligas Al-Xwt%Si-5wt%Zn tixoconformadas, Dissertação de M.Sc., FEM/UNICAMP, Campinas, SP, Brasil, 2012.

[20] PERES, G., Avaliação da tixoconformabilidade de ligas Al-Xwt%Si-5wt%Zn, Dissertação de M.Sc., FEM/UNICAMP, Campinas, SP, Brasil, 2012.

[21] PAES, M., *Obtenção e Caracterização de Novas Ligas Al-Si-Mg para Tixoconformação*, Tese de D.Sc., FEM/UNICAMP, Campinas, SP, Brasil, 2004.